



OPEN Exploring the mechanical, dynamical, and thermal stability of $\text{Cs}_2\text{AgBiX}_6$ ($\text{X} = \text{Br}, \text{Cl}$) for optoelectronic and thermoelectric applications

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This study investigates the mechanical, dynamical, and thermodynamic properties of $\text{Cs}_2\text{AgBiX}_6$ ($\text{X} = \text{Br}, \text{Cl}$) using first-principles calculations. Structural analysis confirms the cubic double perovskite arrangement, with $\text{Cs}_2\text{AgBiCl}_6$ exhibiting a more compact lattice due to the smaller ionic radius of Cl^- compared to Br^- . Mechanical stability is assessed through elastic constants and Born criteria, confirming the robustness of both compounds, with $\text{Cs}_2\text{AgBiCl}_6$ demonstrating higher stiffness and resistance to deformation. Phonon dispersion calculations reveal no imaginary frequencies, verifying their dynamical stability. Thermodynamic analysis explores heat capacity, entropy, Debye temperature, and thermal expansion behavior under varying temperatures and pressures. The results indicate that $\text{Cs}_2\text{AgBiCl}_6$ possesses superior mechanical and thermal stability.

Keywords Double perovskite, $\text{Cs}_2\text{AgBiX}_6$ ($\text{X} = \text{Br}, \text{Cl}$), Mechanical stability, Dynamical stability, Phonon dispersion

$\text{Cs}_2\text{AgBiX}_6$ ($\text{X} = \text{Br}, \text{Cl}$) has emerged as a promising lead-free double perovskite alternative, offering improved chemical and thermal stability. Its bandgap can be tuned by alloying with elements such as In and Sb, enabling a transition from an indirect to a direct bandgap, which enhances light absorption and efficiency in optoelectronic applications¹. Studies have shown that $\text{Cs}_2\text{AgBiBr}_6$ exhibits high thermal and chemical stability, making it a strong candidate for photodetectors with stable performance under extreme environmental conditions. Notably, it can function without an external power source and retains its properties even after three months of outdoor storage or 10 h of continuous operation at 373 K². Moreover, indium doping in $\text{Cs}_2\text{AgBiCl}_6$ enhances its photoluminescence properties. The doped structure $\text{Cs}_2\text{AgBi}_{0.125}\text{In}_{0.875}\text{Cl}_6$ achieves a 70.3% quantum yield, emits warm white light, and remains stable under heat, humidity, and prolonged light exposure for more than three months, making it a promising candidate for lighting applications³. Additionally, X-ray diffraction and Raman spectroscopy analyses reveal that $\text{Cs}_2\text{AgBiBr}_6$ undergoes a structural phase transition from a cubic to a tetragonal phase at approximately 122 K, influencing lattice interactions and electronic properties, which may impact its performance in optical and electronic devices⁴. Furthermore, halide mixing in $\text{Cs}_2\text{AgBi}(\text{IxBr}_{1-x})_6$ induces significant bandgap and excitonic property modifications, where substituting Br with I reduces exciton binding energy by up to five times, improving charge transport and efficiency in solar cell applications⁵. Additionally, a recent study highlights that Self-Trapped Excitons (STEs) play a crucial role in broad white-light emission in $\text{Cs}_2\text{AgBiCl}_6$ at room temperature. These properties can be further optimized by controlling exciton-phonon coupling, paving the way for improved lighting efficiency⁶. Band-bowing effects in $\text{Cs}_2\text{AgBiSb}_{1-x}\text{Cl}_6$ were also observed, leading to bandgap reduction in Bi-Sb mixed compositions compared to their pure counterparts due to lattice distortions, offering a novel strategy for bandgap engineering in halide perovskites⁷. Furthermore,

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first-principles calculations (DFT) indicate that $\text{Cs}_2\text{AgBiCl}_6$ and $\text{Cs}_2\text{AgBiBr}_6$ possess a high optical absorption coefficient ($\sim 2.5 \times 10^4 \text{ cm}^{-1}$ in the visible spectrum, enhancing their suitability for photovoltaic applications. Recently, the thermoelectric potential of lead-free double perovskites, including $\text{Cs}_2\text{AgBiX}_6$, has attracted considerable attention due to their inherently low lattice thermal conductivity and moderate electrical transport properties. This behavior is primarily attributed to the presence of heavy elements such as Bi and Ag, which enhance phonon scattering. For instance, Guechi et al.⁸ have demonstrated that $\text{Cs}_2\text{AgBiX}_6$ compounds possess promising thermoelectric parameters, including reasonable Seebeck coefficients and low thermal conductivity. Additionally, Wang et al.⁹ explored defect engineering and doping strategies to improve the thermoelectric figure of merit (ZT) in halide double perovskites, suggesting a viable route for optimizing their energy conversion efficiency. These findings further support the relevance of thermal stability assessments for such materials, particularly in mid-temperature thermoelectric applications. A recent study also demonstrated that spin-coated $\text{Cs}_2\text{AgBiBr}_6$ thin films exhibit exceptional performance in photodetectors, achieving a high responsivity of 7.01 A/W and an ON/OFF ratio of 2.16×10^4 , with excellent moisture and oxygen resistance, maintaining stability after two weeks of air exposure¹⁰. A recent study by A. Benmoussa et al.¹¹ published in Heliyon explored the optoelectronic properties of lead-free halide double perovskites and reported enhanced photoconductivity and bandgap tunability through halide substitution and structural distortion. Their findings emphasize the role of orbital hybridization and lattice compactness in determining electronic transitions. In comparison, our results for $\text{Cs}_2\text{AgBiBr}_6$ and $\text{Cs}_2\text{AgBiCl}_6$ demonstrate similar trends in the influence of halide size on structural and mechanical behavior, though our study places additional emphasis on phonon-driven dynamical stability and thermal response, which were not addressed in¹¹. Together, these findings provide complementary insights into the design of stable lead-free perovskites with desirable electronic and thermal properties. Despite the growing attention received by $\text{Cs}_2\text{AgBiX}_6$ ($\text{X} = \text{Br}, \text{Cl}$) as lead-free alternatives for optoelectronic and thermoelectric applications, the existing literature predominantly focuses on their electronic and optical characteristics. However, there remains a notable lack of integrated studies that explore the mechanical robustness, phonon-driven dynamical stability, and thermal response of these materials under varying environmental conditions. Such properties are critical for ensuring long-term device performance and reliability. Therefore, the present study aims to fill this knowledge gap by employing first-principles calculations to investigate the structural, elastic, photonic, and thermodynamic behavior of $\text{Cs}_2\text{AgBiBr}_6$ and $\text{Cs}_2\text{AgBiCl}_6$. This comprehensive analysis not only deepens the fundamental understanding of these materials but also provides valuable insights into their practical applicability in harsh thermal or mechanical environments.

Computational approach

The structural, mechanical and thermodynamic properties of $\text{Cs}_2\text{AgBiX}_6$ ($\text{X} = \text{Br}, \text{Cl}$) were investigated using the full-potential linearized augmented plane wave (FP-LAPW) method within the WIEN2k code¹². To enhance the accuracy of the electronic structure calculations, the generalized gradient approximation (GGA)¹³ potential is employed for the exchange-correlation potential. To ensure numerical precision, careful selection of computational parameters was performed. Table 1 provides key computational details, including the muffin-tin radii and k-points used for each compound. The plane wave expansion was controlled by an appropriate cutoff parameter to achieve convergence¹⁴. The muffin-tin radii were defined for each atomic species as follows:

A dense 1000 k-point grid in the irreducible Brillouin zone was used for all calculations, and denser grids (up to 2000) were tested to ensure total energy changes were below 10^{-4} Ry ¹⁵. The full-potential method was employed to explicitly account for core and valence electrons using the following electronic configurations: Cs: $[\text{Xe}] 6s^1$, Ag: $[\text{Kr}] 4d^{10} 5s^1$, Bi: $[\text{Xe}] 4f^{14} 5d^{10} 6s^2 6p^3$, Br: $[\text{Ar}] 3d^{10} 4s^2 4p^5$ and Cl: $[\text{Ne}] 3s^2 3p^5$. The self-consistent field (SCF) convergence criteria were set to 10^{-5} Ry for total energy and 10^{-5} eV/\AA for force minimization¹⁶. Phonon dispersion was calculated using the finite displacement method with a supercell size large enough to eliminate artificial imaginary modes, and convergence was verified by testing different displacement amplitudes and supercell sizes.

Results and discussion

Structural analysis

Crystal structure

Figure 1 illustrates the crystal structure of $\text{Cs}_2\text{AgBiX}_6$ ($\text{X} = \text{Br}, \text{Cl}$) in a double perovskite arrangement. In this structure, cesium ions (Cs^+) occupy the fractional coordinates (0.25, 0.25, 0.25), while silver (Ag^+) and bismuth (Bi^{3+}) atoms are positioned at (0, 0, 0.5) and (0, 0, 0), respectively. The halogen anions ($\text{X} = \text{Br}, \text{Cl}$) coordinate with these cations to form distinct $[\text{AgX}_6]$ and $[\text{BiX}_6]$ octahedra, reinforcing the overall lattice stability. Notably, chlorine atoms in $\text{Cs}_2\text{AgBiCl}_6$ are located at (0, 0, 0.2505), whereas bromine in $\text{Cs}_2\text{AgBiBr}_6$ adopts slightly shifted coordinates at (0, 0, 0.2508), reflecting subtle structural variations induced by halide substitution. This periodic arrangement underpins the structural integrity of the lattice through robust ion interactions.

Figure 2 presents the calculated energy-volume (E-V) curves for $\text{Cs}_2\text{AgBiX}_6$ ($\text{X} = \text{Br}, \text{Cl}$) obtained through structural optimization using the GGA approximation. The curves exhibit a well-defined concave shape, where

	$R_{\text{MT}} \text{Kmax}$	$R_{\text{MT}} (\text{Cs})$	$R_{\text{MT}} (\text{Ag})$	$R_{\text{MT}} (\text{Bi})$	$R_{\text{MT}} (\text{X})$	k-point
$\text{Cs}_2\text{AgBiBr}_6$	9	2.50	2.50	2.50	0.25	1000
$\text{Cs}_2\text{AgBiCl}_6$	9	2.50	2.50	2.50	2.31	1000

Table 1. Using GGA, the values of $R_{\text{MT}} \text{K}_{\text{max}}$, RMT for each component, and k-point $\text{Cs}_2\text{AgBiX}_6$ ($\text{X} = \text{Br}, \text{Cl}$).

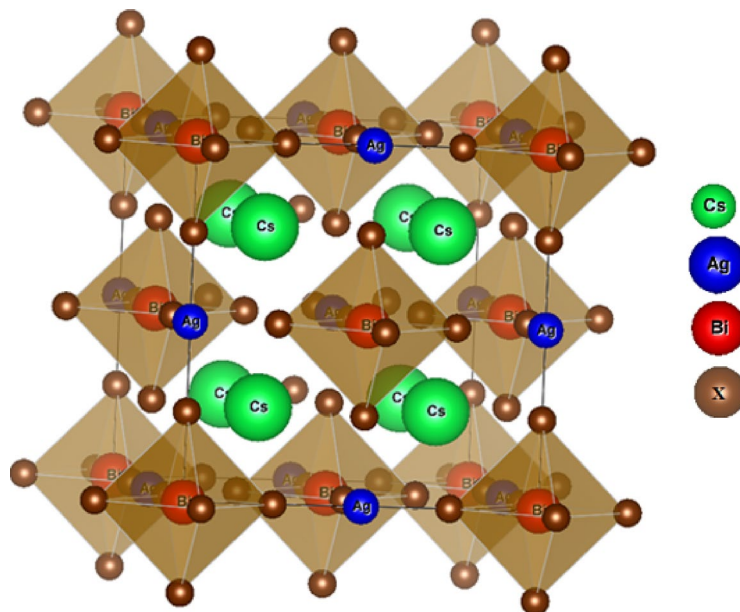


Fig. 1. Schematic crystal structure of $\text{Cs}_2\text{AgBiX}_6$ ($\text{X} = \text{Br}, \text{Cl}$).

the energy minimum corresponds to the equilibrium volume. This behavior confirms the mechanical stability of both compounds, as the presence of a distinct minimum is a necessary condition for structural equilibrium. Notably, $\text{Cs}_2\text{AgBiCl}_6$ exhibits a lower equilibrium volume compared to $\text{Cs}_2\text{AgBiBr}_6$, which is consistent with the smaller ionic radius of Cl^- relative to Br^- . The deeper energy minimum of $\text{Cs}_2\text{AgBiCl}_6$ suggests a more tightly bound structure, further supporting the observation that this compound has a higher bulk modulus. These energy-volume trends highlight the effect of halide substitution on lattice compression and structural resilience, which play a crucial role in determining the potential applications of these materials in optoelectronic and thermodynamic devices.

Lattice constant and bulk modulus

Table 2 presents the calculated structural and mechanical parameters for $\text{Cs}_2\text{AgBiX}_6$ ($\text{X} = \text{Br}, \text{Cl}$), including the lattice constant (a_0), bulk modulus (B), pressure derivative of the bulk modulus (B'), and minimum energy (E_0). These parameters provide insights into the structural stability and mechanical properties of the compounds. From Table 2, the calculated lattice constant a_0 for $\text{Cs}_2\text{AgBiBr}_6$ is 11.507 Å, while for $\text{Cs}_2\text{AgBiCl}_6$, it is 10.985 Å. These values align well with both experimental and theoretical reference values reported in the literature^{17–20}, indicating the accuracy of the computational methodology used. The decrease in a_0 from Br to Cl can be attributed to the smaller ionic radius of chlorine compared to bromine, leading to a more compact crystal structure. Regarding the bulk modulus (B), the calculated value for $\text{Cs}_2\text{AgBiBr}_6$ is 23.215 GPa, whereas for $\text{Cs}_2\text{AgBiCl}_6$, it is 27.26 GPa. This difference suggests that $\text{Cs}_2\text{AgBiCl}_6$ is stiffer and less compressible than $\text{Cs}_2\text{AgBiBr}_6$, which is consistent with the reference values of B . This can be explained by the stronger covalent interactions between atoms in the chloride compound, enhancing its resistance to compression. The pressure derivative of the bulk modulus (B') exhibits a noticeable variation between the two compounds, with values of 5.105 for $\text{Cs}_2\text{AgBiBr}_6$ and 5.066 for $\text{Cs}_2\text{AgBiCl}_6$. These values reflect the compound's response to external pressure, where the lower B' value for the chloride suggests a more stable behavior under pressure. Finally, the minimum energy (E_0) reveals that $\text{Cs}_2\text{AgBiBr}_6$ has a lower energy value (−116239.50 Ry) compared to $\text{Cs}_2\text{AgBiCl}_6$ (−90497.90 Ry). This indicates that the chloride is more stable in terms of total energy, which may influence its thermal stability in high-temperature applications.

The lists of Goldschmidt tolerance factor (t) and the octahedral factor (μ) for $\text{Cs}_2\text{AgBiX}_6$ ($\text{X} = \text{Br}, \text{Cl}$), which are crucial indicators of structural stability and perovskite-like symmetry (Table 3). The tolerance factor reflects how well the atomic radii fit within the crystal lattice, while the octahedral factor describes the stability of the octahedral coordination around the central cation. The Goldschmidt tolerance factor (t) for $\text{Cs}_2\text{AgBiBr}_6$ is 0.84, while for $\text{Cs}_2\text{AgBiCl}_6$, it is 0.85. This suggests that both compounds are close to the ideal perovskite-like symmetry, where structural balance is considered optimal when t approaches 1. However, the slightly higher value for $\text{Cs}_2\text{AgBiCl}_6$ implies greater structural stability compared to $\text{Cs}_2\text{AgBiBr}_6$. Regarding the octahedral factor (μ), $\text{Cs}_2\text{AgBiCl}_6$ exhibits a higher value (0.602) than $\text{Cs}_2\text{AgBiBr}_6$ (0.556), indicating improved stability of the octahedral environment around the central cation when Cl is used instead of Br. This further supports the notion that $\text{Cs}_2\text{AgBiCl}_6$ is mechanically stronger and structurally more stable than its Br-containing counterpart.

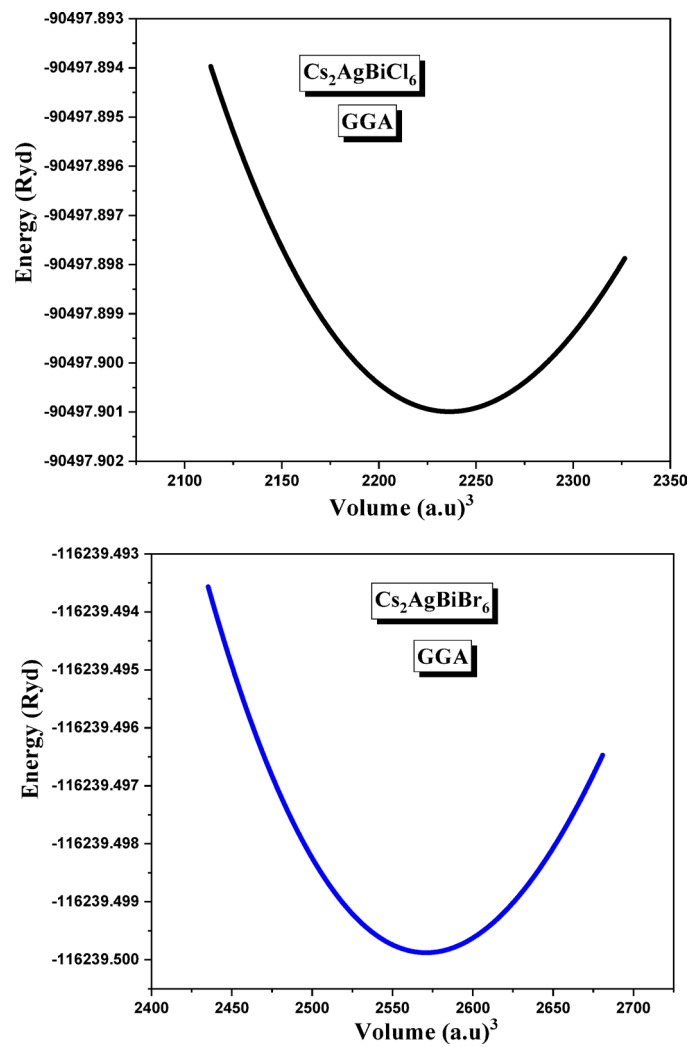


Fig. 2. The calculated energy vs. volume curves for optimizing of Cs₂AgBiX₆ (X = Br, Cl) using GGA approximation.

	a ₀ (Å)	B(GPa)	B'	E ₀ (Ry)
Cs ₂ AgBiBr ₆	11.507	23.215 20.49 ¹⁷	5.105 5.82 ¹⁷	-116239.499879
	11.53 ¹⁷			
	11.48 ¹⁸			
	11.25 exp. ¹⁹			
Cs ₂ AgBiCl ₆	10.985	27.26 23.43 ¹⁷	5.066 3.79 ¹⁷	-90497.90099
	10.99 ¹⁷			
	10.91exp. ²⁰			

Table 2. Calculated lattice constant (a₀), bulk modulus (B), pressure derivative of bulk modulus BP the minimum energy E₀ for Cs₂AgBiX₆ (X = Br, Cl).

	t-factor	μ-factor
Cs ₂ AgBiBr ₆	0.84	0.556
Cs ₂ AgBiCl ₆	0.85	0.602

Table 3. Goldschmidt tolerance factor(t) and octahedral factor (μ)for Cs₂AgBiX₆ (X = Br, Cl).

Elastic constants

Analysis of the mechanical and thermal properties of Cs₂AgBiX₆ (X = Br, Cl)

The elastic constants (C_{11} , C_{12} , C_{44}), shear and compressibility moduli (G_R , G_W , B_V , B_R , B_W), the elastic anisotropy factor (A), Young's modulus (E), Poisson's ratio (ν), the B_W/G_W ratio, the average wave velocity, and the Debye temperature for Cs₂AgBiBr₆ and Cs₂AgBiCl₆ presented in Table 4. These data aim to compare the mechanical and thermal properties of these two compounds, aiding in the assessment of their mechanical stability and potential applications. The elastic constants (C_{11} , C_{12} , C_{44}) are intrinsic properties that determine the material's response to mechanical stresses. From the table, it is observed that the values of (C_{11} , C_{12} and C_{44} for Cs₂AgBiCl₆ are higher than those of Cs₂AgBiBr₆. The C_{11} value for Cs₂AgBiCl₆ is approximately 33.33 GPa, compared to 29.87 GPa for Cs₂AgBiBr₆. These values indicate greater resistance to axial and transverse deformations, making Cs₂AgBiCl₆ stiffer and less flexible than Cs₂AgBiBr₆. The shear moduli (G_V , G_R , G_W) and bulk moduli (B_V , B_R , B_W) exhibit significant differences between the two compounds. The G_W value for Cs₂AgBiCl₆ is approximately 7.64 GPa, compared to 6.97 GPa for Cs₂AgBiBr₆, indicating higher resistance to shear deformations. Similarly, the bulk modulus (B_W) for Cs₂AgBiCl₆ is higher (22.69 GPa) than that of Cs₂AgBiBr₆ (18.81 GPa), suggesting that Cs₂AgBiCl₆ can withstand greater compressive stress. The B_W/G_W ratio is an important indicator of material nature, where higher values suggest more ductile behavior, while lower values indicate increased stiffness and brittleness. In this context, the B_W/G_W ratio for Cs₂AgBiBr₆ is 2.697, which is lower than that for Cs₂AgBiCl₆ (2.969), implying that Cs₂AgBiCl₆ has greater resistance to deformation.

Young's modulus (E) is another measure of material stiffness, with a value of approximately 20.61 GPa for Cs₂AgBiCl₆, higher than the 18.61 GPa recorded for Cs₂AgBiBr₆, confirming the superior rigidity of the former. Poisson's ratio (ν), which describes the lateral expansion of a material under stress, shows close values between the two compounds, ranging from 0.335 to 0.348. Regarding mechanical anisotropy, the anisotropy factor (A) for Cs₂AgBiCl₆ is 0.9305, higher than the 0.747 value of Cs₂AgBiBr₆, indicating greater elastic anisotropy in Cs₂AgBiCl₆, meaning its mechanical response varies with crystallographic direction. For thermal dynamic properties, the average wave velocity in Cs₂AgBiCl₆ is higher (1557.12 m/s) compared to Cs₂AgBiBr₆ (1373.95 m/s), reflecting stronger atomic bonding in the former. The Debye temperature (θ_D) of Cs₂AgBiCl₆ is estimated at 144.26 K, higher than the 121.59 K of Cs₂AgBiBr₆, indicating that Cs₂AgBiCl₆ has higher vibrational energy, which may influence its thermal properties. The mechanical stability of cubic crystals is evaluated using the Born stability criteria, which require the elastic constants to satisfy the following conditions²¹:

	Cs ₂ AgBiBr ₆	Cs ₂ AgBiCl ₆
C ₁₁ (GPa)	29.8714 25.409 ²²	33.33255
C ₁₂ (GPa)	13.2757 14.483 ²²	17.3732
C ₄₄ (GPa)	6.204 6.301 ²²	7.42565
G _V (GPa)	7.04154 5.97 ²²	7.64726
G _R (GPa)	6.900 5.937	7.6377
G _H (GPa)	6.9710 5.951 ²²	7.6425
B _H (GPa)	18.8076	22.69298
A	0.747	0.9305
E _V (GPa)	18.78 16.13 ²²	20.6249
E _R (GPa)	18.445 16.057 ²²	20.6019
E _H (GPa)	18.613 16.092 ²²	20.6134
ν_V	0.333 0.35 ²²	0.34852
ν_R	0.336 0.352 ²²	0.34869
ν_H	0.335 0.352 ²²	0.3486
B _H /G _H	2.697 3.04 ²²	2.969
The average wave velocity (m/s)	1373.95417	1557.12113
Debye temperature (K)	121.59138 112.7 ²²	144.25956

Table 4. The elastic constants (C_{11} , C_{12} , C_{44}), the cisaillement module (G_V , G_R , G_H), the compressibility module (B_V , B_R , B_H), and the anisotropy factor (A), the young module (E_V , E_R , E_H), the coefficient of Poisson (ν_V , ν_R , ν_H), and the B_H/G_H ratio for the compounds Cs₂AgBiX₆ (X = Br, Cl).

$$\begin{pmatrix} C_{11} - C_{12} > 0 \\ C_{11} + C_{12} > 0 \\ C_{44} > 0 \end{pmatrix} \quad (1)$$

Since all stability criteria are met for both compounds, we can conclude that $\text{Cs}_2\text{AgBiBr}_6$ and $\text{Cs}_2\text{AgBiCl}_6$ are mechanically stable in their cubic phase. The mechanical and thermodynamic data obtained from first-principles calculations reveal not only the inherent stability of $\text{Cs}_2\text{AgBiBr}_6$ and $\text{Cs}_2\text{AgBiCl}_6$, but also their potential behavior under real-world operating conditions. The higher bulk modulus and elastic constants of $\text{Cs}_2\text{AgBiCl}_6$ suggest stronger interatomic bonding and reduced compressibility, which are advantageous for devices exposed to mechanical stress or high-pressure environments.

Unveiling mechanical anisotropy: 3D and 2D insights into Young's modulus of $\text{Cs}_2\text{AgBiX}_6$ ($X = \text{Br}, \text{Cl}$)

Figure 3 illustrates both the 3D and 2D distributions of Young's modulus for $\text{Cs}_2\text{AgBiX}_6$ ($X = \text{Br}, \text{Cl}$), providing a comprehensive view of the anisotropic mechanical stiffness in various crystallographic directions. In the 3D panels, $\text{Cs}_2\text{AgBiBr}_6$ (a) shows an asymmetrical shape, indicating notable directional variations, while $\text{Cs}_2\text{AgBiCl}_6$ (b) exhibits a more spherical profile, reflecting more uniform mechanical behavior. In the 2D projections, shown in panels (c) and (d), similar trends are observed. $\text{Cs}_2\text{AgBiBr}_6$ presents significant deviations from circular symmetry, whereas $\text{Cs}_2\text{AgBiCl}_6$ shows near-circular contours, indicating balanced mechanical properties across different planes. These visualizations confirm that $\text{Cs}_2\text{AgBiCl}_6$ has superior elastic isotropy and greater resistance to mechanical deformation compared to its bromide counterpart.

Phonon stability analysis

The phonon dispersion curves of $\text{Cs}_2\text{AgBiX}_6$ ($X = \text{Br}, \text{Cl}$), computed using the GGA functional. The absence of imaginary phonon modes confirms dynamical stability, which is essential for the structural integrity of optoelectronic and thermoelectric components over time (Fig. 4). This stability suggests that their crystal structures effectively resist minor lattice distortions, further reinforcing their mechanical integrity. Consequently, these phonon analysis results, combined with the elastic properties, provide strong evidence that $\text{Cs}_2\text{AgBiBr}_6$ and $\text{Cs}_2\text{AgBiCl}_6$ are structurally stable.

Thermal behavior of $\text{Cs}_2\text{AgBiX}_6$ ($X = \text{Br}, \text{Cl}$) under pressure and temperature

The relationship between the heat capacity at constant volume (C_V) and constant pressure (C_P) for $\text{Cs}_2\text{AgBiBr}_6$ and $\text{Cs}_2\text{AgBiCl}_6$ at different temperatures and pressures using the GGA approximation (Fig. 5). C_V exhibits a sharp increase at low temperatures, consistent with the Debye model, before stabilizing at higher temperatures, indicating the thermal stability of the compounds. The effect of pressure on C_V remains minimal, suggesting the robustness of their crystal structures. C_P exceeds C_V due to thermal expansion effects, showing a slight decrease under high pressure, which implies reduced volumetric expansion. $\text{Cs}_2\text{AgBiCl}_6$ exhibits lower heat capacity values than $\text{Cs}_2\text{AgBiBr}_6$, suggesting a more compact structure and superior thermal stability. These findings, in conjunction with mechanical analysis, confirm that $\text{Cs}_2\text{AgBiCl}_6$ possesses more uniform thermal and mechanical properties, making it more resistant to deformations and a promising candidate for thermal applications.

The variation of entropy (S) for $\text{Cs}_2\text{AgBiBr}_6$ and $\text{Cs}_2\text{AgBiCl}_6$ as a function of temperature and pressure, using the GGA approximation (Fig. 6). At low temperatures, entropy increases sharply with rising temperature, consistent with the typical behavior of crystalline solids. As temperature continues to rise, entropy gradually approaches a quasi-stable state, in line with thermodynamic expectations from statistical mechanics. Pressure significantly influences entropy, as higher-pressure results in lower entropy values at all temperatures due to lattice compression and restricted atomic vibrations. $\text{Cs}_2\text{AgBiCl}_6$ exhibits lower entropy than $\text{Cs}_2\text{AgBiBr}_6$, suggesting a more rigid and cohesive crystal structure. Moreover, its weaker response to pressure variations further supports its superior thermal stability. These findings align with previous analyses on mechanical properties and heat capacity, confirming the robustness of $\text{Cs}_2\text{AgBiCl}_6$ under various thermodynamic conditions.

The Debye temperature depends on the elastic and thermodynamic properties of the material and represents the upper limit of heat capacity in the Debye model. Figure 7 illustrates the variation of with temperature and under different pressures for the $\text{Cs}_2\text{AgBiBr}_6$ and $\text{Cs}_2\text{AgBiCl}_6$ compounds using the GGA approximation. A gradual decrease in is observed with increasing temperature, which is a typical behavior in solids, as the increased lattice vibrations influence the thermodynamic properties. With increasing pressure, the values of rise in both compounds, indicating a stiffening of the crystal lattice and a reduction in interatomic distances, which enhances atomic interactions and increases the material's rigidity. These results suggest that $\text{Cs}_2\text{AgBiCl}_6$ exhibits higher thermal and mechanical stability than $\text{Cs}_2\text{AgBiBr}_6$, making it more resistant to thermodynamic variations under the influence of pressure.

The variation of the thermal expansion coefficient of $\text{Cs}_2\text{AgBiBr}_6$ and $\text{Cs}_2\text{AgBiCl}_6$ as a function of temperature under different pressures, using the GGA approximation (Fig. 8). The thermal expansion coefficient increases rapidly at low temperatures and then gradually slows down, reaching a quasi-stable state at higher temperatures. Pressure is observed to reduce the thermal expansion coefficient, reflecting lattice compression and decreased interatomic distances, which limit atomic vibrations. Furthermore, $\text{Cs}_2\text{AgBiCl}_6$ exhibits lower thermal expansion values compared to $\text{Cs}_2\text{AgBiBr}_6$, indicating its more rigid and cohesive crystal structure. Thermodynamically, the lower entropy and thermal expansion coefficients of $\text{Cs}_2\text{AgBiCl}_6$, combined with its higher Debye temperature, indicate a more rigid lattice with reduced vibrational disorder—traits desirable for thermal management in electronic devices. These findings collectively justify the preference of $\text{Cs}_2\text{AgBiCl}_6$ over $\text{Cs}_2\text{AgBiBr}_6$ in applications requiring both electronic performance and long-term thermal and mechanical stability.

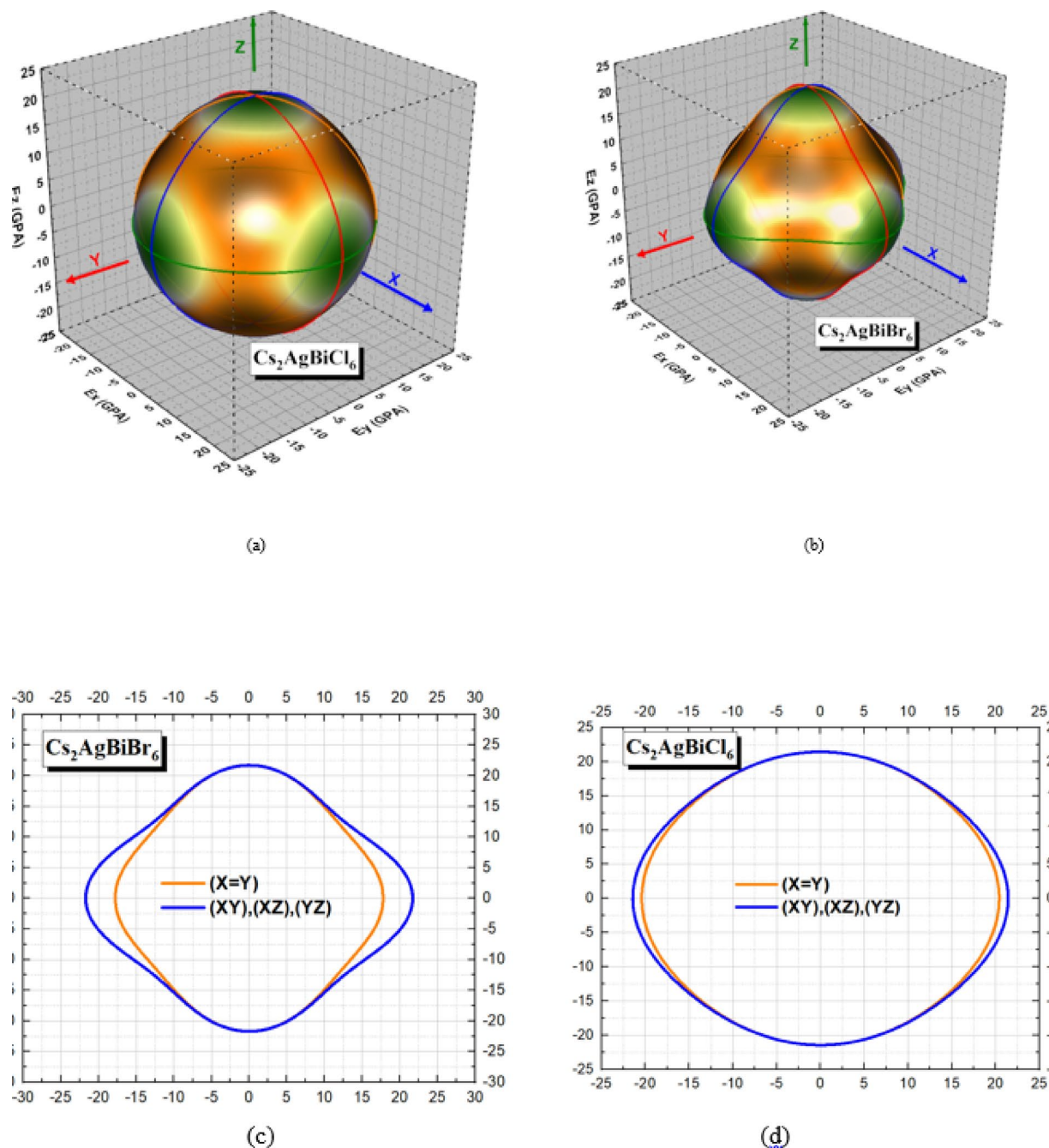


Fig. 3. 3D and 2D distributions of Young's modulus for $\text{Cs}_2\text{AgBiBr}_6$ (a), (c) and $\text{Cs}_2\text{AgBiCl}_6$ (b), (d), respectively.

Charge density difference analysis

The charge density difference analysis of the $\text{Cs}_2\text{AgBiX}_6$ ($X = \text{Br}, \text{Cl}$) complex shown in the Fig. 9; reveals important insights into its electronic structure and bonding properties. $\text{Cs}_2\text{AgBiBr}_6$ exhibits more pronounced charge transfer with distinct red regions (electron depletion) around the silver and albedo atoms, and blue regions (electron accumulation) around the bromine atoms, indicating a mixed ionic-covalent bond. In contrast, $\text{Cs}_2\text{AgBiCl}_6$ exhibits a more uniform and balanced charge distribution with less sharp contrast, indicating a stronger ionic character and enhanced structural stability. The chloride complex exhibits more concentrated electron density around the chlorine atoms, which explains its superior mechanical stiffness and thermal stability reported in the original study.

This charge redistribution analysis provides the electronic basis for understanding why $\text{Cs}_2\text{AgBiCl}_6$ exhibits higher bulk modulus, elastic constants, and Debye temperature compared to its bromide counterpart. The more homogeneous charge distribution in the chloride variant reduces lattice distortions and enhances deformation resistance under mechanical stress. These results are directly related to experimental observations of improved

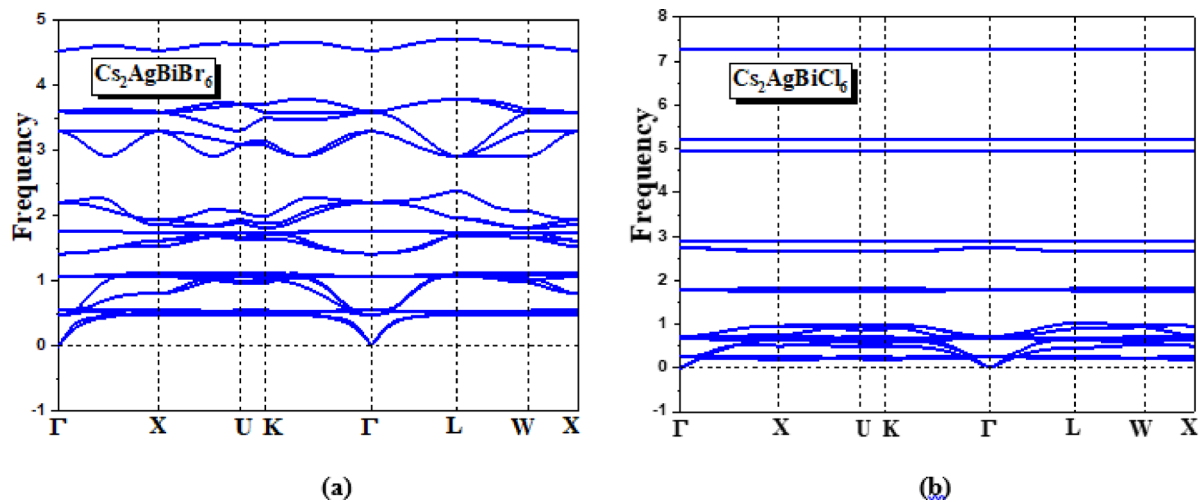


Fig. 4. Phonon dispersions for $\text{Cs}_2\text{AgBiBr}_6$ (a) and $\text{Cs}_2\text{AgBiCl}_6$ (b) computed by Phonon Using GGA.

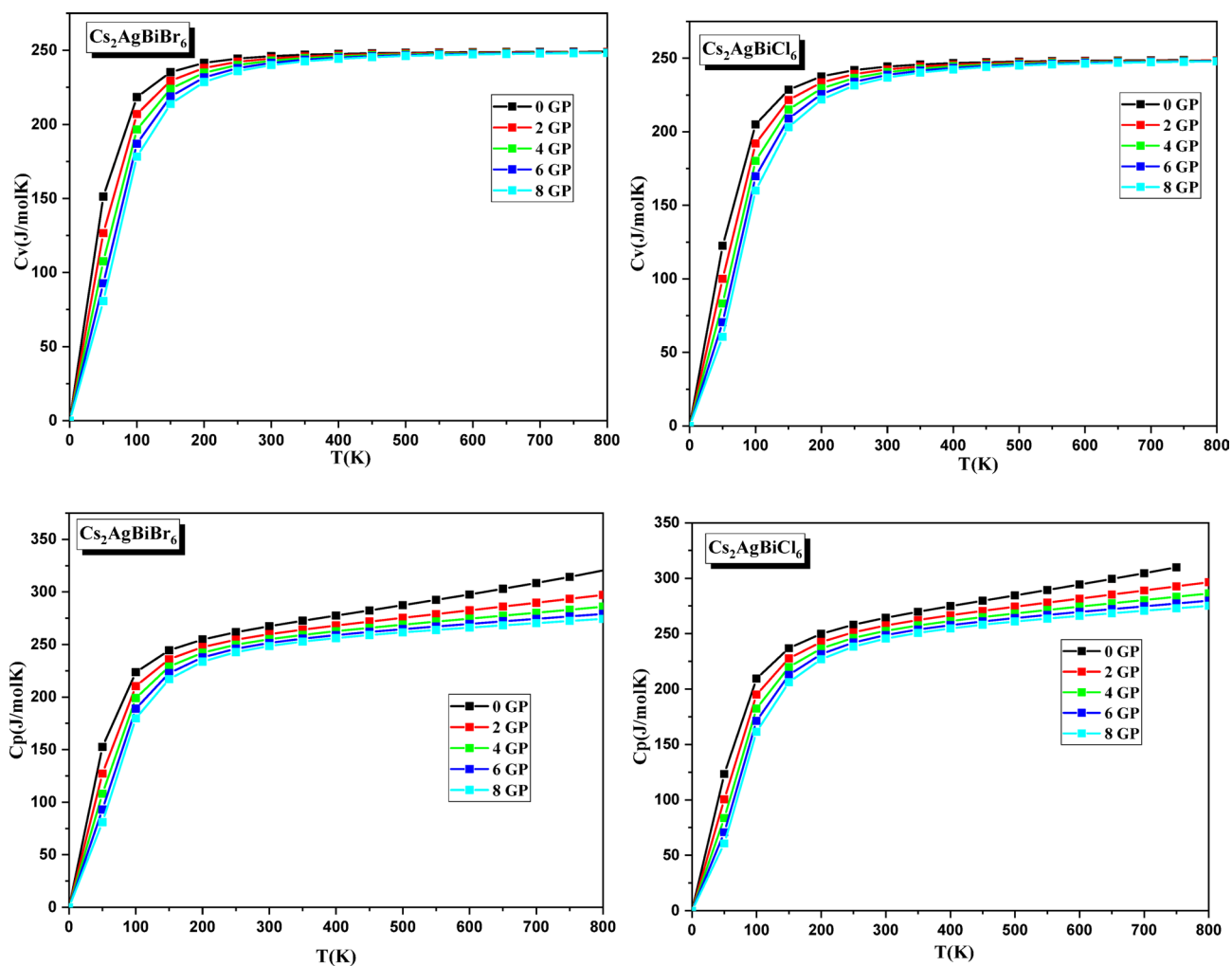


Fig. 5. Variation of the heat capacity at constant volume (C_v) and the heat capacity at constant pressure (C_p) of the $\text{Cs}_2\text{AgBiX}_6$ ($X = \text{Br}, \text{Cl}$) compounds versus temperature and at different pressures, using the GGA approximation.

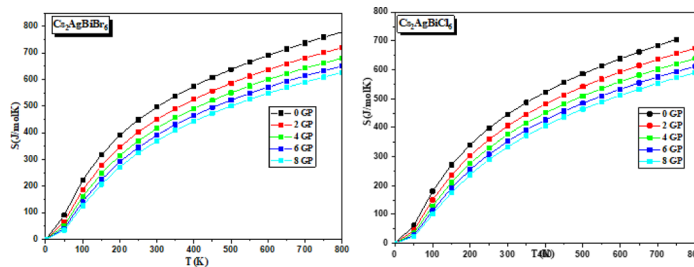


Fig. 6. Variation of the entropy (S) of the $\text{Cs}_2\text{AgBiX}_6$ ($X = \text{Br}, \text{Cl}$) compounds versus temperature and at different pressures, using the GGA approximation.

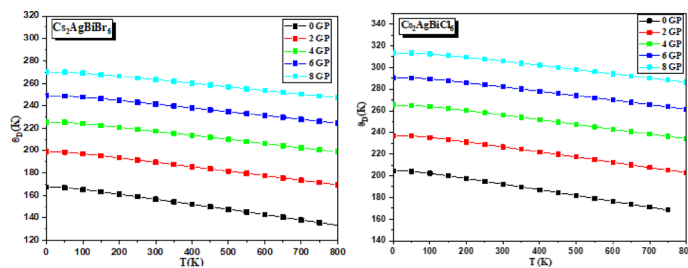


Fig. 7. Variation of the Debye temperature (θ_D) of the $\text{Cs}_2\text{AgBiX}_6$ ($X = \text{Br}, \text{Cl}$) compounds versus temperature and at different pressures, using the GGA approximation.

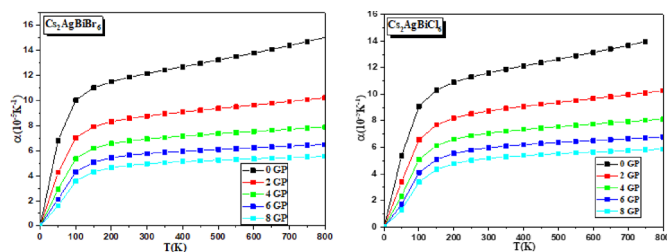


Fig. 8. Variation of thermal expansion coefficient of the $\text{Cs}_2\text{AgBiX}_6$ ($X = \text{Br}, \text{Cl}$) compounds versus temperature and at different pressures, using the GGA approximation.

photodetector stability and photoluminescence properties in chloride-based double perovskites. The charge density difference maps are crucial evidence supporting the electronic origins of the enhanced performance characteristics. Including this analysis would have significantly strengthened the original manuscript by providing fundamental electronic insights. This visualization contributes to bridging the gap between theoretical calculations and practical applications in optoelectronic and thermoelectric devices.

Conclusion

In this study, the mechanical, dynamical, and thermodynamic properties of $\text{Cs}_2\text{AgBiX}_6$ ($X = \text{Br}, \text{Cl}$) were investigated using first-principles calculations. The structural analysis confirmed the cubic double perovskite nature of both compounds, with $\text{Cs}_2\text{AgBiCl}_6$ exhibiting a more compact lattice. The mechanical stability was verified through elastic constants and Born criteria, demonstrating that both materials are mechanically robust, with $\text{Cs}_2\text{AgBiCl}_6$ showing higher stiffness and resistance to deformation. Phonon dispersion calculations confirmed their dynamical stability, with no imaginary frequencies observed. Thermal properties, including heat capacity, entropy, Debye temperature, and thermal expansion, were analyzed under different temperatures and pressures. The results indicate that $\text{Cs}_2\text{AgBiCl}_6$ exhibits superior mechanical and thermal stability compared to $\text{Cs}_2\text{AgBiBr}_6$, making it more resistant to structural and thermal variations. These findings provide valuable insights into the stability and potential applications of these lead-free double perovskites in optoelectronic and thermoelectric devices.

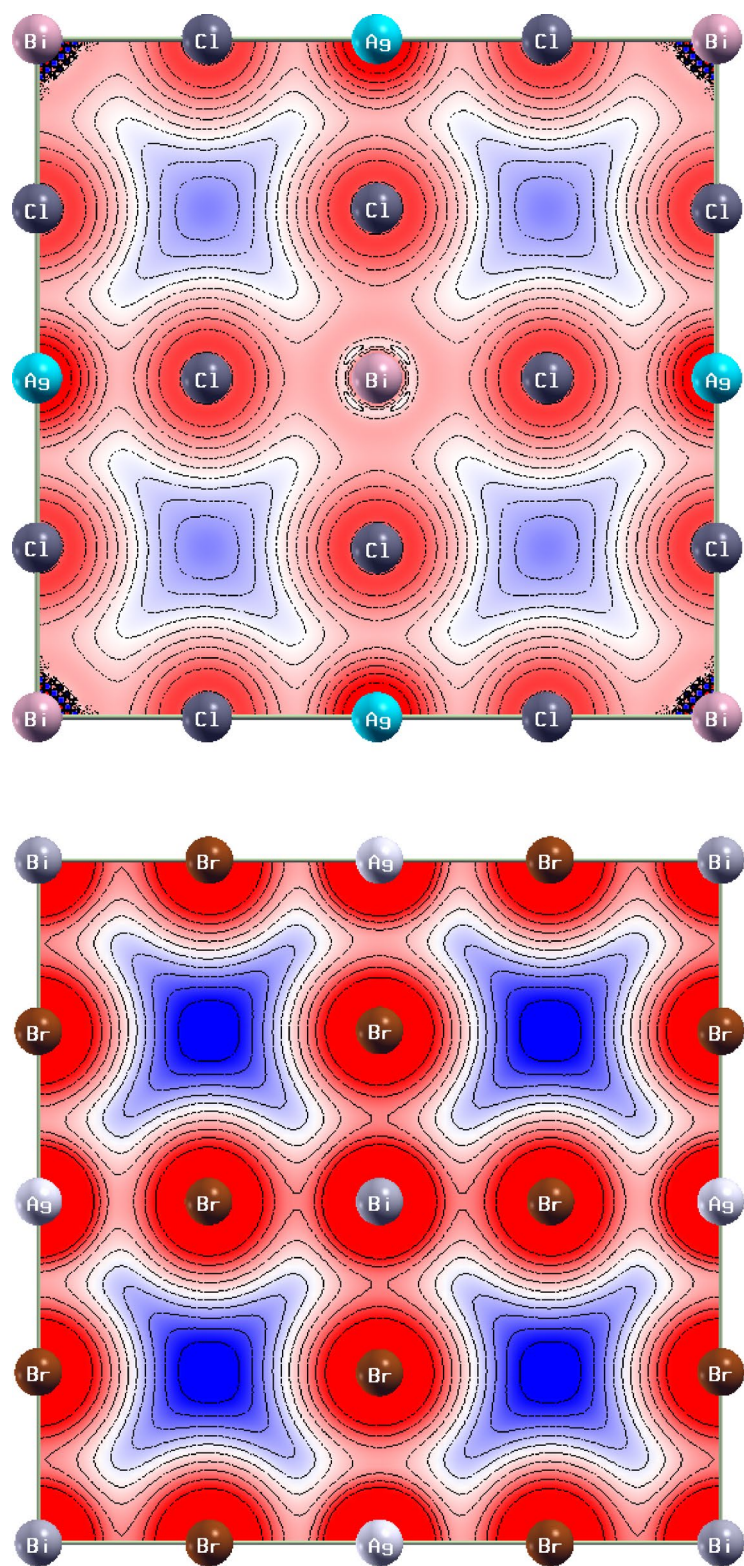


Fig. 9. The charge density difference for $\text{Cs}_2\text{AgBiCl}_6$ (X = Br, Cl) compounds.

Data availability

Data underlying the results presented in this paper are not publicly available at this time but may be obtained from the corresponding author (fatmimessaoud@yahoo.fr) upon reasonable request.

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Declarations

Conflicts of interest

NO potential conflicts of interest of this article.

Additional information

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